

Quantum-chemical study of excitons in tetragonal BaTiO₃ and SrTiO₃ crystals

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ABSTRACT

Using a quantum-chemical INDO method based on the Hartree-Fock formalism and the periodic large unit cell (LUC) model we present a theoretical interpretation of the structural and electronic properties of triplet excitons in the tetragonal BaTiO₃ and SrTiO₃ crystals. Our study demonstrates that the exciton structure has particularities in each material. In the BaTiO₃ the defect structure corresponds to the so-called Mott-Wannier-type exciton having a considerable separation, 7.0 Å, between the hole and the electron. Meanwhile, in SrTiO₃ the structural and electronic features of the triplet exciton are quite different. The hole-electron distance is about 2.14 Å and the defect is well localized in two contiguous atoms: the hole on one of the O atoms and the electron on the neighbor Ti atom. The calculated luminescence energy using the so-called Δ SCF method is found to be equal to 0.94 eV and 1.13 eV for BaTiO₃ and SrTiO₃, respectively. Since it falls within the infrared part of the spectrum, the experimentally detected green luminescence due to photo-excited states should be attributed to the singlet excitons.

Keywords: BaTiO₃, SrTiO₃, excitons, LUC, structural and electronic properties, luminescence

1. INTRODUCTION

Barium titanate (BaTiO₃) and strontium titanate (SrTiO₃) are the most important perovskite-type titanates due to their wide range of applications as electronic materials. The primitive unit cell of BaTiO₃ and SrTiO₃ contain five atoms¹ similar to other perovskite-type crystals. Although both titanates have similar structure, they exhibit different physical, chemical and electronic properties. Technologically, these differences yield in use of these two materials in different applications. Among a number of important features, BaTiO₃ has large electro-optic coefficients and high photorefractive sensitivity.² This material also can be used in self-pumped phase conjugation or holographic storage³ and inexpensive diode lasers.⁴ Other important properties are its chemical and mechanical stability in a wide temperature range, which facilitates its fabrication in bulk polycrystals and both epitaxial and polycrystalline thin films.⁵ Meanwhile, SrTiO₃ is used as a grain-boundary barrier layer capacitor, oxygen-gas sensor, epitaxial growth substrate for high-temperature superconductor thin films and catalytic material. SrTiO₃ has also been known to become superconducting when a small number of electron carriers are doped by La substitution for a Sr atom⁶ (see references therein).

The structures of both barium and strontium titanates have simple cubic symmetry, which transform to the tetragonal ones below the characteristic transition temperature of the material. For BaTiO₃ the transition occurs below 380 K while SrTiO₃ does transform to tetragonal phase below 105 K. This corresponds to the transition: paraelectric phase \rightarrow ferroelectric phase. The static dielectric constant increases extraordinarily when the temperature decreases, e.g. SrTiO₃ is stabilized at over 10^4 below 10 K without a ferroelectric phase transition.⁷ This property is known as quantum paraelectricity (QPE). The nature of QPE has not yet been revealed.⁸ Despite the fact that many studies have been done so far on these ferroelectric materials, there is still no sufficient understanding of the nature of many properties and phenomena in both barium and strontium titanates. On the other hand, it is well understood that point defects play an important role in many applications of BaTiO₃ and SrTiO₃ crystals and therefore, further studies are necessary on this subject. One of the most common defects in non-metallic crystals is the so-called exciton. Free excitons created by exciting radiation undergo transformation into self-trapped excitons in singlet or triplet states. Later on the recombination between electrons and trapped holes causes the

appearance of the recombination luminescence of the self-trapped exciton. Some considerable efforts have been done to understand the nature of excitons in many oxides. As an example we can note that the adiabatic potential surface for triplet exciton self-trapping has been studied in the SiO₂ crystal⁹ using the same computer code as we do. The results interpreted in great detail the nature of the self-trapped-exciton optical absorption and luminescence in the SiO₂ crystal.⁹

There are not many studies on excitons in perovskites due to the complexity of possible spatial configurations of the exciton in these materials and necessity to take into consideration rather complex chemical bonding, which includes also a covalent component. However, recently quantum-chemical calculations on self-trapped excitons in KTaO₃ and KNbO₃ crystals have been performed¹⁰ using again the same computer code. Thus, it has been showed repeatedly that the CLUSTERD computer code¹¹ is adequate for the exciton studies in different oxides. Additionally, we have tested the present method in a number of other applications on titanates. Some examples include studies of hole self-trapping in a pure¹² and defective BaTiO₃ crystals,¹³ hole self-trapping studies in the SrTiO₃ cubic lattice,^{14,15} structural and electronic features of triplet excitons in the tetragonal BaTiO₃ crystal¹⁶ and defect doping studies in the bulk of SrTiO₃,^{6,17,18} CaTiO₃,¹⁹ BaTiO₃,^{20,21} PbTiO₃²² crystals as well as quantum-chemical computations of defects on the SrTiO₃ surfaces.^{23,24}

The purpose of the present article is to obtain equilibrium geometry and electronic structure of the triplet exciton in both BaTiO₃ and SrTiO₃ tetragonal lattices. Since these materials have different physical, chemical and electronic properties, the geometry and the electronic structure of triplet excitons in these crystals are expected to be different. We also study the lattice distortion due to the exciton and possible luminescence due to the hole-electron pair annihilation.

2. OUTLINE OF THE COMPUTATIONAL METHOD

The employed method of periodically repeated large unit cells (LUCs)⁹ is designed for the calculation of the total energy

Table 1. Numerical parameter sets used in the present work: ζ , E_{neg} , P^0 and β .

Atom	AO	ζ (au)	E_{neg} (eV)	P^0 (e)	β (eV)
Ba	6s	1.65	11.34	0.2	-0.4
	5p	2.8	30.6	2.0	-4.0
Ti	4s	1.4	1.4	0.65	-0.5
	4p	1.1	-10.0	0.04	-0.5
	3d	1.93	-2.9	0.62	-9.0
O	2s	2.27	4.5	1.974	-16.0
	2p	1.86	-12.6	1.96	-16.0
Sr	5s	1.59	11.15	0.18	-0.4
	4p	2.80	37.5	2.00	-4.5

and electronic structure of crystals. The calculations are based on the Hartree-Fock formalism in the framework of the intermediate neglect of differential overlap (INDO) approach. We chose the LUC model because of its advantage compared to different cluster models, for example, better treatment of the exchange interaction. Additionally, it is important to note that from the viewpoint of simulating the exciton self-trapping, the original equivalency of the perfect lattice sites is essential. Only the periodic models meet this condition. A full discussion of the mathematical equations to calculate the total energy within the framework of the LUC approach is given in numerous works^{9,11,19} carried out before. Here we shall only state that the method includes a number of the so-called semi-empirical parameters. The atomic parameter sets are optimized by calculating the following properties and comparing them to the experimental results: (i) the main features of a given crystal (width of the forbidden energy gap, widths of the upper and lower valence bands, lattice constants of the

crystal), (ii) the inter-atomic distances and ground states of some selected diatomic molecules, and finally (iii) the ionization potentials of the host atoms of the crystal. The numerical parameters for the BaTiO₃ and SrTiO₃ crystals were obtained before reproducing the most important properties of this material and also in studies of other crystals.

Specifically, the numerical parameter sets for the Ba, O and Ti atoms used in the present work were taken from Ref. 20 while the parameters for Sr 5s and 4p AOs from Ref. 18. Thus, we have restricted ourselves to use the so-called valence basis set. The INDO parameters utilized in the present work are given in Tables 1 and 2. Throughout the present work we have exploited a LUC of 135 atoms, i.e., a 27 times (3x3x3) extended primitive unit cell has been used for both barium and strontium titanates. In order to answer the question of the exciton possible localization/delocalization we have performed the total spin density analysis.

Table 2. Electron-core interaction parameters $\alpha_{\mu B}$ for the tetragonal phase. The μ th AO belongs to atom A, where $A \neq B$.

A	$\alpha_{\mu B}$ (au ⁻¹)			
	B = Ba	B = Ti	B = O	B = Sr
Ba	0.20	0.52	0.33	-
Ti	0.11	0.13	0.10	0.10
O	0.52	0.36	0.15	0.59
Sr	-	0.55	0.25	0.15

3. RESULTS AND DISCUSSION

3.1. Structural and electronic features of triplet excitons in the tetragonal BaTiO₃ crystal

As a result of the spin density analysis for the 135 atom LUC, the electronic part of the exciton is found to be well localized on the Ti(11) atom while the hole part settles down mainly on O(8) atom (Fig. 1). The elements of the self-consistent-field (SCF) spin density matrix for the atoms situated within the defective region are given in Table 3. As it follows from these data, 94% of the electron is localized on the 3d_{xy} AO of the Ti(11) atom while the hole part is mainly found, 84% of localization, on the 2p_x AO of the O(8) atom. However, the 2p_x AOs of the O(7) and O(9) share also some part of the hole. This particular localization of the hole may correspond to hole resonance structure occurring among the O(7)-O(8)-O(9) atoms. The approximate distance between the two centers is about 7.0 Å. This implies that we have the so-called Mott-Wannier-type exciton. The exciton is weakly bound, which is a commonly accepted exciton model for the ionic crystals.

The performed geometry optimization leads to a rather strong electronic density re-distribution due to the exciton presence. It is found that the two O(8)-nearest Ti atoms, Ti(5) and Ti(6), situated along the y-axis move outward the O(8) atom by 0.08 Å. The four O(8)-nearest Ba atoms also move outward the defect along the <101> direction by about 0.03 Å. Finally, the two Ti-closest O atoms located along the y-axis, O(7) and O(9), move towards the O(8) atom and thus reduce the chemical bond lengths between Ti(5)-O(7) and Ti(6)-O(9). Considering these particular atomic movements, we have obvious lattice polarization due to the hole presence. This polarization occurs along the y direction and also leads to weakening of the chemical bonding between the oxygen, which receives a hole, and its surrounding cations. It is important to state that the polarization axis does not coincide with the ferroelectric z-axis. Thus one can see that exciton presence in the ferroelectric tetragonal BaTiO₃ phase might reduce ferroelectric polarization in this material and can induce the loss of bulk ferroelectricity in the material.

The analysis of the electronic part of the exciton shows that it is well localized on the Ti(11) atom. Again, due to the vicinity polarization effect we observe the atomic movements along the z-axis as shown in Fig. 1. In the present case the lattice polarization is along the ferroelectric axis. However, its direction is opposite, i.e., the induced local polarization due to the

Fig. 1. The hole part, O(8) atom, and the electron part, Ti(11) atom, of the exciton and displacements of the neighboring atoms. The positive displacements are along the z axis. One can observe the lattice some polarization along the z axis.

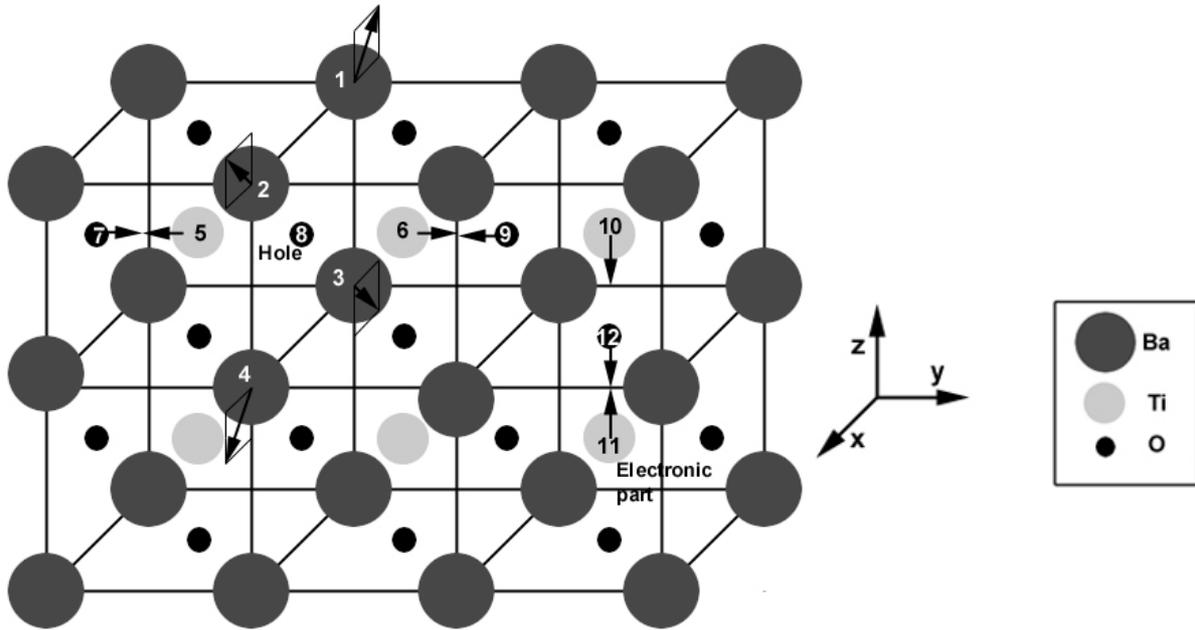


Table 3. SCF spin density matrix elements for the atoms within the defective region. Predominantly, the electronic part is localized on the $3d_{xy}$ AO of the Ti(11) atom while the hole part is found to be localized mainly on the $2p_y$ AO of the O(8) atom. The atomic numeration corresponds to the one given in Fig. 1.

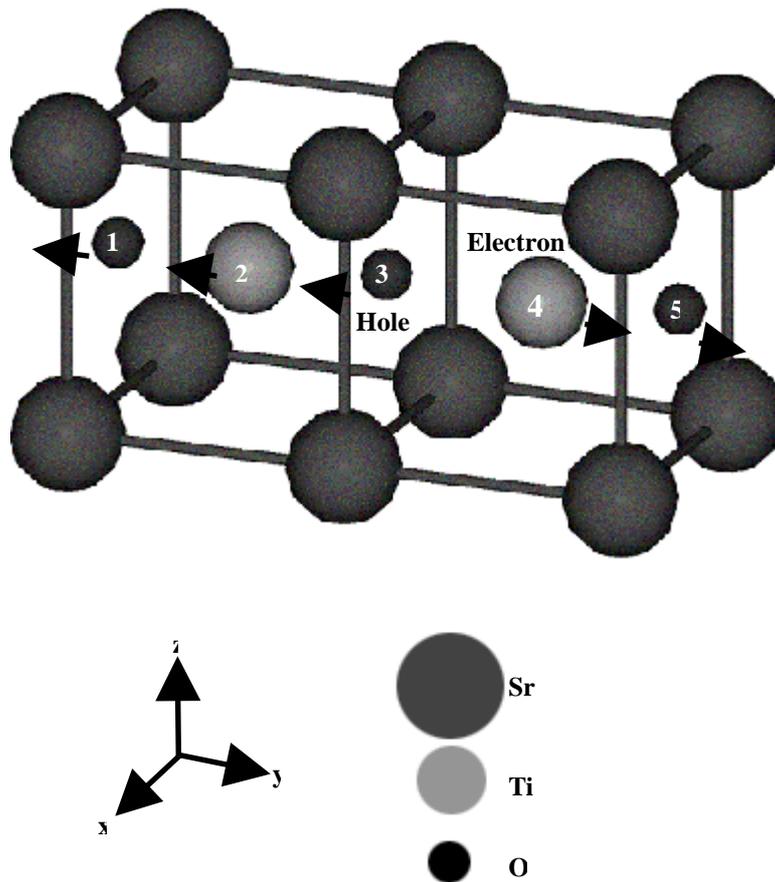
Electron	4s	4p _x	4p _y	4p _z	3d _{x²-y²}	3d _{xy}	3d _{xz}	3d _{yz}	3d _{z²}
Ti(5)	-0.04467	-0.00122	-0.00387	-0.00129	-0.14112	-0.01274	-0.00521	-0.01584	-0.05855
Ti(6)	-0.04467	-0.00122	-0.00387	-0.00129	-0.14112	-0.01274	-0.00521	-0.01584	-0.05855
Ti(11)	0.01551	0.00268	0.0026	0.00085	0.01772	0.93938	0.01282	0.01228	0.03135
		Hole	2s	2p _x	2p _y	2p _z			
		O(7)	0.00394	0.00764	0.27763	0.00921			
		O(8)	0.0042	0.0052	0.85785	0.00512			
		O(9)	0.00394	0.00764	0.27763	0.00921			

electron trapping at the Ti(11) atoms reduces the total ferroelectric moment. The calculated relaxation energy as a difference between the total energies of relaxed and unrelaxed systems is found to be equal to 4.4 eV.

3.2. Structural and electronic features of triplet excitons in the tetragonal SrTiO₃ crystal

As a result of the spin density analysis for the 135 atom LUC, the electronic part of the exciton is found to be well localized on the Ti(4) atom while the hole part settles down on the O(3) atom (Fig. 2). The elements of the SCF spin density matrix for the atoms situated within the defective region are given in Table 4. As it follows from these data 93% of the electron is localized on the 3d_{xy} AO of the Ti(4) atom while the hole part mainly is found, 84% of localization, on the 2p_x (39%) and 2p_y (45%) AOs of the O(3) atom.

Fig. 2. Spatial structure of the triplet exciton in the tetragonal SrTiO₃ lattice. The hole and the electronic parts are localized on the O(3) and Ti(4) atoms, respectively. The surrounding atoms move outwards the defect along the y axis creating the crystalline lattice polarization along the same axis.



The performed geometry optimization as it is shown in Table 5 leads to rather strong electronic density re-distribution. In general, it is found that the movements of five atoms in the defective region are non-negligible. Two of them move up to 4% (0.15 Å) of the magnitude of the corresponding lattice parameter $a_o = 3.88$ Å. It should be mentioned that due to the fact that all atomic movements are along the y axis, we express the lattice distortion in terms of the a_o lattice parameter. The defect-nearest O(1), Ti(2) and O(5) atoms, situated along the y-axis, move outwards the defect (see Fig. 2). As one can see from the atomic charge analysis (Table 5) and also considering the above-mentioned particular atomic movements, there exists obvious lattice polarization due to the hole presence. This polarization occurs along the y direction and also leads to a weakening of the chemical bonding between the oxygen, which receives a hole and its surrounding cations. It is important to state that the polarization axis does not coincide with the ferroelectric z-axis similarly as in the case of the BaTiO₃ crystal discussed above.

The computed distance between the two centers, hole and electron, is found to be approximately 2.14 Å. This implies that the triplet state exciton in the SrTiO₃ crystal is stronger bound compared to the case of the BaTiO₃ crystal. Finally, we can state that the calculated relaxation energy is found to be equal to 3.3 eV.

Table 4. SCF spin density matrix elements for the atoms within the defective region. Predominantly, the electronic part of triplet exciton is localized on the 3d_{xy} AO of the Ti(4) atom while the hole part is found to be localized on the 2p_x and 2p_y AOs of the O(3) atom. The atomic numeration corresponds to the one given in Fig. 2.

Electron	4s	4p _x	4p _y	4p _z	3d _{x²-y²}	3d _{xy}	3d _{xz}	3d _{yz}	3d _{z²}
Ti(2)	-0.0396	-0.0054	-0.0046	-0.0014	-0.0809	-0.0480	-0.0092	-0.0109	-0.0376
Ti(4)	0.0053	0.0016	0.0006	0.0003	0.0056	0.9307	0.0134	0.0103	0.0263
		Hole	2s	2p _x	2p _y	2p _z			
		O(1)	0.0006	0.0058	0.0649	0.0036			
		O(3)	0.0044	0.3868	0.4550	0.0090			
		O(5)	-0.0003	0.0000	0.0222	-0.0032			

Table 5. Atomic displacements, Δr, and changes in atomic charge, Δq, due to the triplet exciton self-trapping in the tetragonal SrTiO₃ lattice. The atomic numeration and the direction of their movements correspond to the one given in Fig. 2.

Atom	Δr (% a _o)	Δq (e)
O(1)	1.0	+0.096
Ti(2)	4.0	-0.280
O(3)	4.0	+0.450
Ti(4)	2.1	-0.500
O(5)	2.0	+0.097

3.3. Luminescence due to the exciton in SrTiO₃ and BaTiO₃

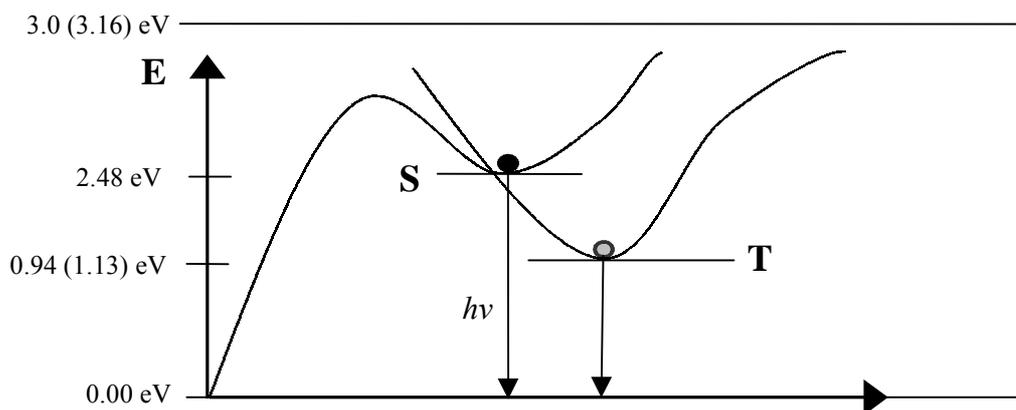
Annihilation of an exciton, i.e., electron and hole mutual recombination gives rise to the luminescence, which can be observed experimentally. In our study, the luminescence energy was obtained as a difference of total energies for the self-consistent-field ground and excited states for optimized triplet exciton geometry (ΔSCF method). That is, the potential energy curve was calculated for the triplet exciton, and then according to the Frank-Condon principle the de-excitation energy is that for the vertical transition from the minimum of the triplet exciton relaxed state to the SCF singlet state of the exciton for a fixed atomic arrangement. Thus computed luminescence energy was found to be equal to 0.94 eV for BaTiO₃ and 1.13 eV for SrTiO₃.

In this connection, some recent results reported for the triplet exciton state in the cubic phase of ABO₃ oxides^{10,25} should be mentioned. The authors propose the charge-transfer vibronic exciton (CTVE) approach to describe the electronic excitations in these systems. However, they use the electronic density matrix in their analysis of electron and hole polaron localization within the crystalline lattice, rather than the SCF spin density matrix, which gives a more precise pattern for the exciton localization. Secondly, the geometry used in their works apparently does not correspond to the equilibrium geometry

leading to incorrect values of calculated luminescence energies. Therefore, the CTVE existence in titanates is rather doubtful.

Our theoretical interpretation of the results obtained in the present work is given in Fig. 3. In our mind, the experimentally observed green luminescence in a number of perovskites is due to singlet excitons in these materials. Since the triplet state

Fig. 3. Total energy curve as a function of the generalized coordinate, Q , for the self-trapped exciton in the tetragonal BaTiO_3 and SrTiO_3 lattices (energy values in parenthesis correspond to SrTiO_3). The luminescence of the singlet state might be responsible for the experimentally detected green light, while the triplet exciton state lies considerably lower and as a result might participate in the photodegradation of the material.



of a given excited system normally lies lower than the singlet state, the luminescence energy due to the triplet exciton should be considerably smaller than the band-gap width. According to the experimental data, the band-gap width in BaTiO_3 and SrTiO_3 crystals is around 3.0 eV¹⁶ and 3.16 eV,⁸ respectively. The observed green luminescence of 2.48 eV⁸ in SrTiO_3 is very close to this magnitude. Therefore, we expect the green luminescence to occur due to the singlet exciton annihilation as it is shown in Fig. 2. Meantime, the triplet exciton may contribute to the photodegradation process of the material rather than in the emission of green light since the magnitude of its luminescence energy, 0.94 eV for BaTiO_3 and 1.13 eV for SrTiO_3 , falls within the infrared spectrum.

4. CONCLUSIONS

The developed for crystals quantum-chemical approach has been applied to study the triplet state of exciton in tetragonal BaTiO_3 and SrTiO_3 crystals. The structural, electronic and optical properties of the triplet exciton are investigated in detail for both materials. The obtained geometry and the electronic structure of triplet excitons in these crystals are different. The model of the so-called Mott-Wannier exciton, which normally occurs in ionic crystals, describes firmly our results on the exciton geometry of the BaTiO_3 . However, the triplet exciton in the SrTiO_3 lattice resembles characteristics of the two limiting approximations Frenkel and Mott-Wannier. For the BaTiO_3 , the electron part is well localized on one of the Ti atoms while the hole resonances among three neighboring oxygens. The inter-particle distance is calculated to be equal to 7.0 Å. Meanwhile, in the SrTiO_3 , the electron and the hole are well localized on Ti and O atoms, respectively, with inter-particle distance equal to 2.14 Å. In both perovskite-type crystals, the obtained atomic displacements within the defective region and electronic charge density re-distribution point to a strong crystalline lattice polarization around the defect. Furthermore, in the BaTiO_3 lattice, the exciton-induced dipole moment is opposite to the existing ferroelectric polarization, thus leading to the loss of bulk ferroelectricity in the material. The ΔSCF calculated luminescence energy, 0.94 eV for BaTiO_3 and 1.13 eV for SrTiO_3 , falls within the infrared spectrum. This implies that the experimentally determined luminescence energy of photo-excited state annihilation in the tetragonal SrTiO_3 crystal, 2.48 eV,⁸ apparently corresponds to the de-excitation of singlet state excitons in this material.

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